

Supplementary Materials

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Experimental Section

Materials Characterization: X-ray diffraction (XRD) patterns of the samples were obtained using a multipurpose rotating target X-ray diffractometer with a Cu K α radiation source ($\lambda = 1.54178 \text{ \AA}$). Morphological characterization Scanning Electron Microscope (SEM) and Energy-Dispersive X-ray Spectroscopy (EDS) was performed using an SU8220 cold field emission scanning electron microscope (15 kV, JEOL, JSM-6700F). X-ray photoelectron spectroscopy (XPS) results were acquired from an ESCALAB 250Xi spectrometer. Mo K-edge X-ray absorption spectra were collected in transmission/fluorescence mode at the 1W1B beamline of the Beijing Synchrotron Radiation Facility (BSRF, Beijing). During the measurements, the X-rays were monochromatized by a Si(111) double-crystal monochromator. Additionally, the position of the absorption edge was calibrated using Mo foil. All measurement data for ex situ X-ray absorption fine structure (XAFS) were collected within one period. Spectral analysis was performed using the Athena software^[1].

Electrochemical Measurements: Electrochemical tests were performed using a CH660D electrochemical workstation with a three-electrode system, employing the prepared samples as the working electrode, a carbon rod as the counter electrode, and Ag/AgCl (3M KCl) as the reference electrode. The glassy carbon working electrode was prepared by dispersing 2 mg of the sample in a solution containing 750 μL of water, 250 μL of isopropanol, and 30 μL of Nafion solution, followed by 30 min of sonication. A 10 μL aliquot of the mixed solution was then drop-cast onto the surface of the glassy carbon electrode (3 mm diameter) and dried at 60°C for 2 h. Cyclic voltammetry (CV) curves were recorded at various scan rates, and galvanostatic charge-discharge curves (GCD) were measured at different current densities within a voltage window of -0.1-0.4 V. All electrochemical tests were conducted at room temperature in a 1 M H₂SO₄ electrolyte.

Computational Details: This study jointly uses VASP^[2] and JDFTx^[3]. The former is used for structural optimization and electronic structure calculations, while the latter is utilized to compute properties related to the double layer^[4, 5]. The Vienna Ab initio Simulation Package (VASP) utilizes density functional theory (DFT) calculations

applying the Perdew-Burke-Ernzerhof (PBE) formulation within the generalized gradient approximation (GGA) [6]. Ion cores are characterized using projected augmented wave (PAW) potentials, and a plane wave basis set with a 450 eV kinetic energy cutoff accounts for valence electrons[7]. Gaussian smearing allows for partial occupancies of Kohn-Sham orbitals with a width parameter of 0.05 eV.

Self-consistent electronic energy requires an energy fluctuation less than 0.01 eV/Å. The Brillouin zone employs a 7×4×1 k-point grid, and self-consistent field (SCF) calculations operate with an energy of 1×10^{-8} eV. Convergence in geometry optimization is achieved once the energy change falls below a defined threshold. To obtain a more accurate electronic structure, a $19 \times 11 \times 1$ k-point grid is utilized. A vacuum of at least 15 Å exists perpendicular to the surface in all systems.

JDFTx employs the GGA-PBEsol functional to describe exchange-correlation energy. The ultrasoft pseudopotential characterizes the interaction between atomic nuclei and electrons. The kinetic energy cutoff is defined at 20 Hartree. The implicit electrolyte is modeled by the linear polarizable continuum model (LinearPCM) coupled with the CANDLE model in a 1M NaF solution to capture solvation effects and the electrolyte response to surface net charge[8].

The quantum capacitance is calculated using the fixed-band approximation, as per the following formula[4, 9]:

$$C_Q = \frac{dQ}{d\phi} = \frac{e^2}{4KT} \int_{-\infty}^{+\infty} D(E) \text{sech}^2 \left[\frac{E-\phi}{2kT} \right] dE \quad (1)$$

here, $D(E)$ represents the density of states, and ϕ is the applied potential.

The double-layer potential is calculated by determining the shift in the Fermi energy level within a unit voltage, as per the following formula[10, 11]:

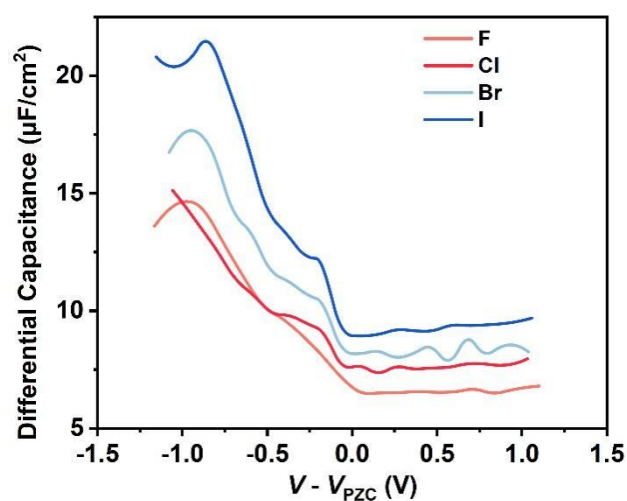
$$C_{EDL} = \Delta\sigma / \Delta E \quad (2)$$

where $\Delta\sigma$ represents the surface charge density and ΔE represents the relative displacement of the Fermi level.

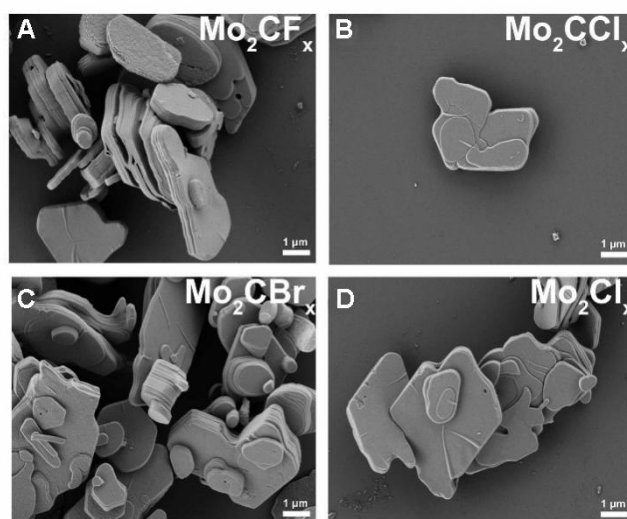
Then, using the same surface charge Q , the corresponding φ_Q and φ_{EDL} are obtained.

In the case of two capacitors in series, $\varphi_{Tot} = \varphi_Q + \varphi_{EDL}$, implying that the total

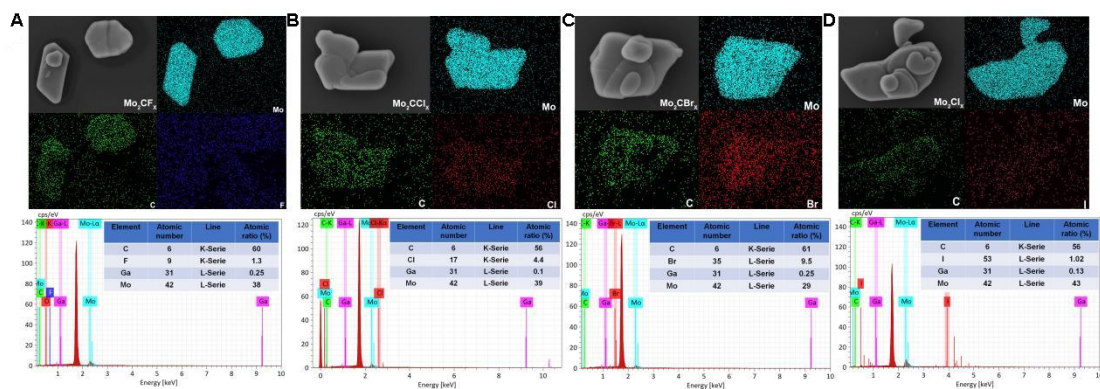
capacitance $C_{Tot} = Q/\varphi_{Tot}$ is equal to $1/C_{Tot} = 1/C_Q + 1/C_{EDL}$.^[11]



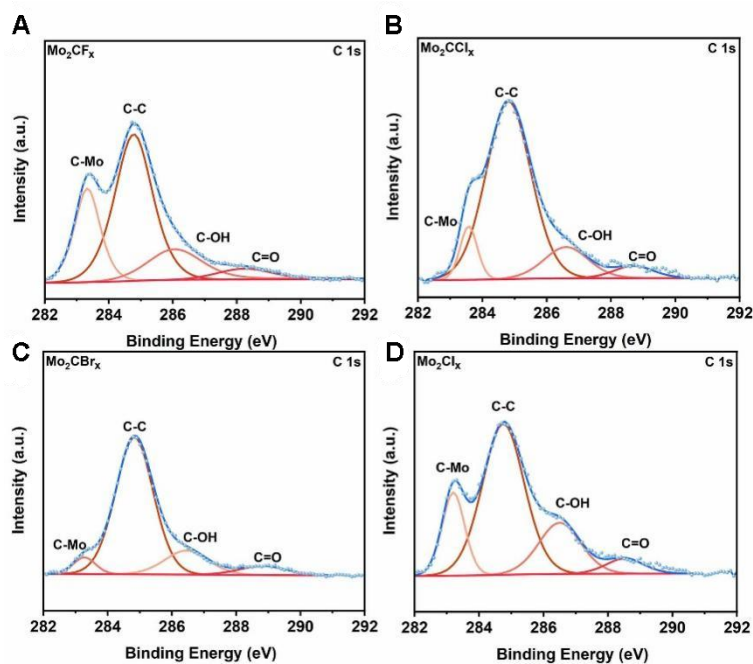
Supplementary Figure 1. Differential capacitance of Mo_2CT_x ($T = \text{F}, \text{Cl}, \text{Br}, \text{I}, \text{O}$) MXene.



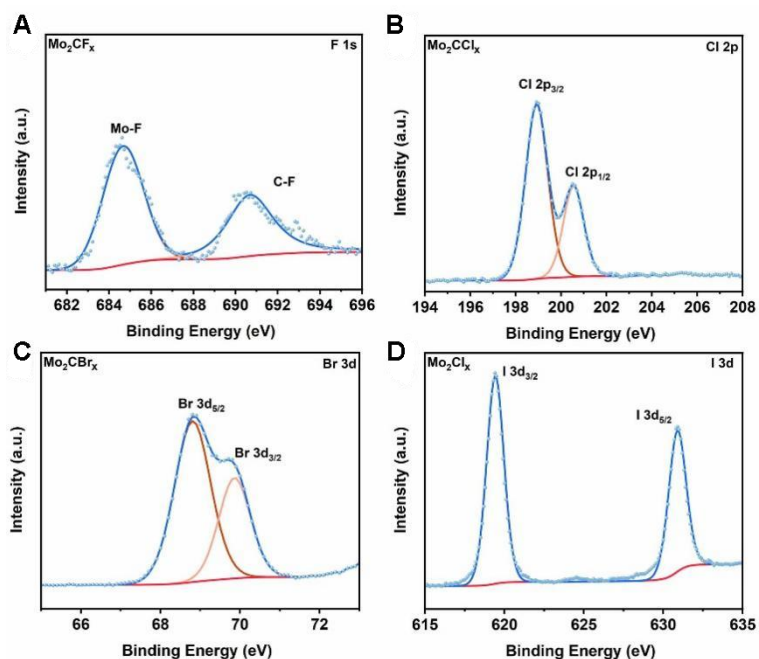
Supplementary Figure 2. SEM images of Mo_2CT_x ($T = \text{F}, \text{Cl}, \text{Br}, \text{I}$). (A) Mo_2CF_x . (B) Mo_2CCl_x . (C) Mo_2CBr_x . (D) Mo_2CI_x .



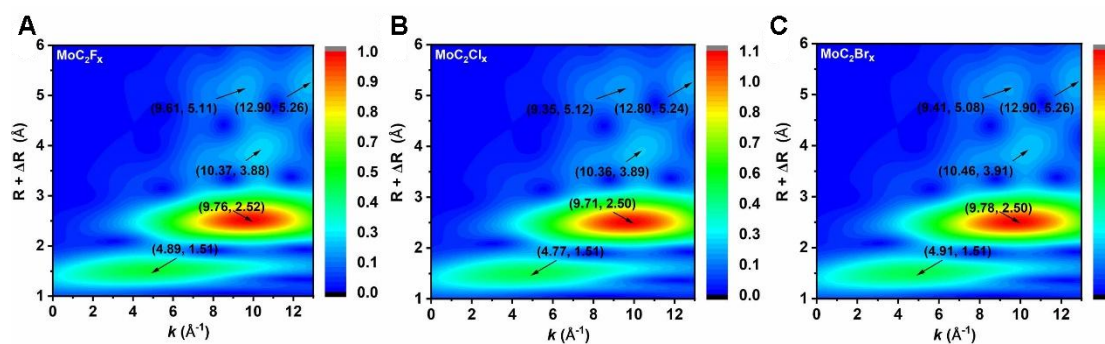
Supplementary Figure 3. EDS elemental distribution analysis on Mo₂CT_x (T = F, Cl, Br, I). (A) Mo₂CF_x. (B) Mo₂CCl_x. (C) Mo₂CBr_x. (D) Mo₂CI_x.



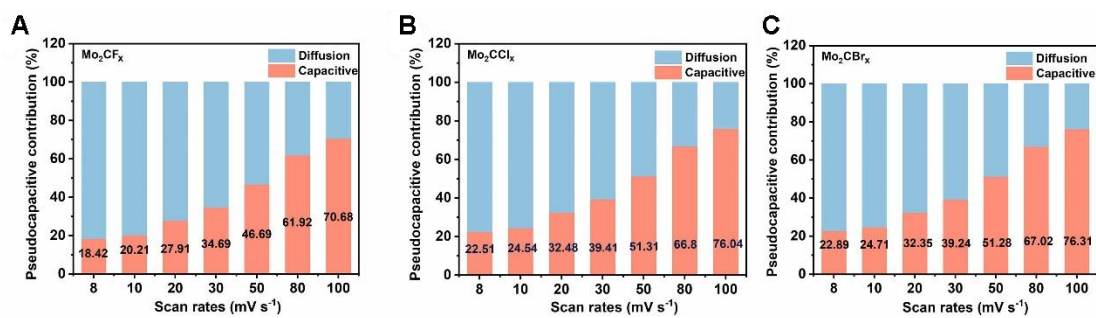
Supplementary Figure 4. XPS spectrum of C 1s for Mo₂CT_x (T = F, Cl, Br, I). (A) Mo₂CF_x. (B) Mo₂CCl_x. (C) Mo₂CBr_x. (D) Mo₂CI_x.



Supplementary Figure 5. XPS spectrum of surface termination for Mo_2CT_x (T = F, Cl, Br, I). (A) F 1s for Mo_2CF_x . (B) Cl 2p for Mo_2CCl_x . (C) Br 3d for Mo_2CBr_x . (D) I 3d for Mo_2CI_x .



Supplementary Figure 6. Wavelet transform (WT) of Mo_2CT_x (T = F, Cl, Br, I). (A) Mo_2CF_x . (B) Mo_2CCl_x . (C) Mo_2CBr_x . (D) Mo_2CI_x .



Supplementary Figure 7. The calculated diffusion and capacitive ratios at different scan rates for Mo_2CT_x ($T = \text{F, Cl, Br}$). (A) Mo_2CF_x . (B) Mo_2CCl_x . (C) Mo_2CBr_x .

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